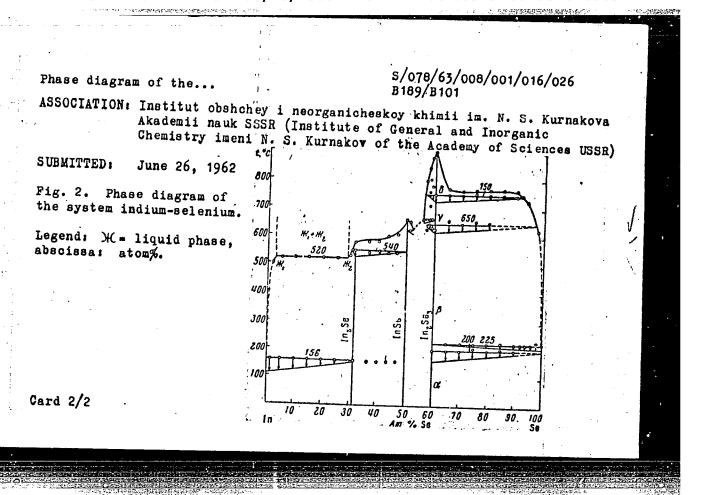
S/078/63/008/001/016/026 B189/B101

AUTHORS: Slavnova, G. K., Luzhnaya, N. P., Medwedeva, Z. S.

TITLE: Phase diagram of the system indium - selenium

PERIODICAL: Zhurnal neorganicheskoy khimii, v. 8, no. 1, 1963, 153 - 159

TEXT: To study the system indium - selenium a series of melts was prepared in quartz ampoules with argon atmosphere, with compositions varying between 98 at% Se + 2 at% In and 2 at% Se + 98 at%. In. The reaction temperatures varied between 600° and 900° C according to the composition of the mixture. The heating time was 6-10 hours. The annealing temperatures were $190\pm10^{\circ}$ C or $400\pm10^{\circ}$ C. The phase diagram (Fig. 2) of the system In-Se was plotted on the basis of the thermal analysis of the samples; in some cases also on that of X-ray analysis, which gave corresponding results. The regions where the known compounds InSe, In₂Se, and In₂Se exist were determined. The following melting points were obtained for these substances: InSe $660\pm10^{\circ}$ C, In₂Se, $900\pm10^{\circ}$ C, In₂Se $540\pm10^{\circ}$ C (melting under decomposition). The following temperatures of polymorphous conversions were found: $\alpha \rightleftharpoons \beta$: $200\pm10^{\circ}$ C; $\beta \rightleftharpoons \gamma$: $650\pm10^{\circ}$ C; $\gamma \rightleftharpoons \delta$: $750\pm10^{\circ}$ C.



S/078/63/008/002/007/012 B101/B186

AUTHORS:

Ts'ui Ping-hsin, Luzhnaya, N. P., Konstantinov, V. I.

TITLE:

Investigation of the ternary reciprocal system of potassium and tantalum fluorides and chlorides

PERIODICAL: Zhurnal neorganicheskoy khimii, v. 8, no. 2, 1963, 389 - 395

TEXT: Both the system KF - KCl - K₂TaF₇, which is important for the electrolytic production of tantalum, and the systems K₂TaF₇ - KCl and K₂TaF₇ were investigated. Differing from T. Juchi et al. (Bull. Res. Inst. Mineral Dress. and Metallurgy Tohoku Univ., 15,87 (1959)) it was found in the binary systems that the compounds K₂TaF₇·KCl (N = 1.434) and K₂TaF₇·KF (N = 1.423, N = 1.420, N = 1.416) melt congruently at 776°C and undergo a polymorphic transformation at 741°C. In the system containing KCl two eutectics exist; In the system with KF, the eutectic containing 21.5 mole% KF melts at 717°C, the one containing 74.5 mole% KF at 727°C. The system KF - KCl - K₂TaF₇ may Card 1/3

Investigation of the ternary...

S/078/63/008/002/007/012 B101/B186

be subdivided into two independent secondary systems: The ternary system K₂TaF₇·KF - K₂TaF₇·KCl - K₂TaF₇ and the system corresponding to the exchange reaction KCl + K₂TaF₇·KF - K₅ KF + K₂TaF₇·KCl. For KCl, KF, K₂TaF₇·KCl, K₂TaF₇·KF, α-K₂TaF₇·KF + K₂TaF₇·KCl. For KCl, KF, K₂TaF₇·KCl, K₂TaF₇·KF, α-K₂TaF₇·KF + K₂TaF₇·KCl. For KCl, K₂TaF₇·KCl, K₂TaF₇·KF, α-K₂TaF₇·KCl + Primary crystallization regions were determined. The invariant points have the following position: E₁ at 580°C and 41.5 mole% KF, 51.5 mole%. KCl, 7.0 mole% K₂TaF₇; E₂ at 710°C, 11.6 mole%KF, 8.7 mole% KCl, 79.7 mole%. K₂TaF₇; P at 678°C, 62.0 mole% KF, 19.0 mole% KCl, 19.0 mole% K₂TaF₇. From the partially plotted phase diagram of the system K, Ta||Cl,F it follows that the compound KCl·2KF·TaF₅ (or K₂TaF₇·KCl), melting congruently at 776°C, exists and that the cross section (KCl)₅ - K₂TaF₇ is a stable binary system. From the investigation of the melting-point diagram of KF - KCl - K₂TaF₇·it followed that in the usual electrolyte used for the production of tantalum or Ta-Nb alloys only KCl, KF and K₂TaF₇·KCl exist before Ta₂O₅ or Nb₂O₅ are added, and that no free K₂TaF₇ or K₂TaF₇·KF are Card 2/3

Investigation of the ternary...

S/078/63/008/002/007/012 B101/B186

formed. The data established can be used to select optimum compositions of the electrolyte. There are 7 figures and 2 tables.

ASSOCIATION: Institut obshchey i neorganicheskoy khimii im. N. S. Kurnakova Akademii nauk SSSR (Institute of General and Inorganic Chemistry imeni N. S. Kurnakov of the Academy of Sciences USSR);

Moskovskiy elektrolampovyy zavod (Moscow Electric Lamp Plant)

SUBMITTED: July 4, 1962

Card 3/3

S/078/63/008/002/008/012 B101/B186

AUTHORS:

Ts!ui Ping-hsin, Konstantinov, V. I., Luzhnaya, N. P.

TITLE:

Phase solubility and interaction in systems containing Ta205,

potassium and tantalum fluorides and chlorides

PERIODICAL: Zhurnal neorganicheskoy khimii, v. 8, no. 2, 1963, 396 - 402

TEXT: To clarify the electrochemical processes used for producing tantalum the following partial systems belonging to the quaternary reciprocal system K,Ta||F,Cl,O were investigated: All binary systems of the tetrahedron (KCl)₁₀ - Ta₂O₅ - K₂TaF₇ - (KF)₁₀, except KCl - KF, partially the systems K,Ta||F,Cl and K,Ta||FO and the cross sections K₂TaF₇·KCl - Ta₂O₅ and KCl - Ta₂O₅·2K₂TaF₇. It was found that the solubility of Ta₂O₅ in KCl at 950°C is only 0.04% by weight, whereas the solubility of Ta₂O₅ in KF at 1122°C is 35% by weight. In the system KF - Ta₂O₅ an eutectic was found at 853°C and 1.8% by weight Ta₂O₅. Above this concentration of Ta₂O₅ there occurs a Card 1/3

S/078/63/008/002/008/012 B101/B186

Phase solubility and interaction ...

crystallization not of Ta205 but of primary KTaO3 by way of the reaction 3KF + Ta205 = KTaO3 + K2TaO2F3. In the system K2TaF7 - Ta205 the formation of the compound Ta205.2K2TaF7 was confirmed. Investigation of the interaction between KF, K2TaF7.KCl, KCl, on the one hand and of Ta205, Ta205.2K2TaF7 on the other hand points to the following conclusion: In KCl not even 1% by weight Ta205.2K2TaF7 is soluble at 1000°C, whereas in molten K2TaF7.KCl 15 mole% Ta205 can be dissolved. Ta205.2K2TaF7 crystallizes from the melt according to the reaction 2(K2TaF7.KCl) + Ta205.2K2TaF7 crystallizes from the melt according to the reaction 2(K2TaF7.KCl) + Ta205.2K2TaF7 texton regions of KF, K2TaF7.KF, Ta205.2K2TaF7, K3TaO2F4 and K2TaO2F3, a triple eutecticum at 718°C, 2.0 mole% Ta205.77.5 mole% KF, 20.5 mole% K2TaF7 with KF, K2TaF7KF and K3TaO2F4 crystallizing therein and a peritectic at 764°C, 4.5 mole% Ta205, 87 mole% KF, 8.5 mole% K2TaF7, where K2TaO2F3 goes into solution and KF and K3TaO2F4 crystallize. The existence of the Card 2/3

Phase solubility and interaction...

s/078/63/008/002/008/012 B101/B186

compounds $K_3^{TaO}2^F4$ and $K_2^{TaO}2^F3$ was proved by the shape of the liquidus iso-These were formed as a result of the reactions 2(Ta205.2K2TaF7) + 16KF = $4(K_2^{TaF_7 \cdot KF}) + 5K_3^{TaO_2F_4}$ (3) and $K_3^{TaO_2F_4} = KF + K_2^{TaO_2F_3}$ (4). Accordingly, Ta205 and Ta205.2K2TaF7 are decomposed by KF and form the potassium oxyfluorotantalates $K_n^{TaO_2F_m}$ (n = 1,2,3; m = 3,4). As a result of the reaction (4) the solubility of Ta205.K2TaF7 in the electrolyte reaches 60 mole% at 868°C. Final conclusion: Within the temperature range 750 -850°C of the electrolysis, the compounds K2TaF7 KCl, K3TaO2F4 or K3TaO2F3 exist in the usual electrolyte besides KF and KCl, playing an important role in the electrolytical process. There are 5 figures and 3 tables. ASSOCIATION: Institut obshchey i neorganicheskoy khimii im. N. S. Kurnakova Akademii nauk SSSR (Institute of General and Inorganic Chemistry

imeni N. S. Kurnakov of the Academy of Sciences USSR); Moskovskiy elektrolampovyy zavod (Moscow Electric Lamp Plant) SUBMITTED: July 4, 1962 Card 3/3

LUZHNAYA, N.P. AID Nr. 994.6 20 June

DIAGRAM OF THE In — Se SYSTEM (USSR)

Slavnova, G. K., N. P. Luzhnaya, and Z. S. Medvedeva. Zhurnal neorganicheskoy khimii, v. 8, no. 5, May 1963, 1199-1203.

S/078/63/008/005/011/021

Card 1/2

DIAGRAM OF THE In	Se SYSTEM [Cont'd]	8/078/63/008	/005/011/021	
	at 650 \pm 10°C, and $\Upsilon = \delta$ nell microhardness: In_2 (30 g); α - In_5 Se ₆ , 393 k 20 g). Micrographs of i			
			Card 2/2	

S/0078/64/009/003/0660/0664

ACCESSION NR: AP4019489

AUTHOR: Dembovskiy, S. A.; Luzhnaya, N. P.

TITLE: Phase diagram of the As-Se system

SOURCE: Zhurnal neorg. khimii, v. 9, no. 3, 1964, 660-664

TOPIC TAGS: arsenic selenium system, phase diagram, x ray analysis, differential thermal analysis, As sub 2 Se sub 3, AsSe, As sub 2 Se sub 3-AsSe system, AsSe-As system, Se-As sub 2 Se sub 3 system

ABSTRACT: The phase diagram of the As-Se system was studied by differential thermal and x-ray phase analysis (fig. 1). The melts in the glass-forming area (from Se to about 60 at. % As) were crystallized beforehand by prolonged annealing. Two compounds were found in the system: As₂Se₃, known before, and AsSe, detected by systematic investigation. As₂Se₃ has a sharp maximum, and that of AsSe is leveled. The area of first crystallization of As₂Se₃ is in a wide range of compositions from 20-47 at. % As; for AsSe the range is narrow, from

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ACCESSION NR: AP4019489

47-55 at. % As. Individual diagrams of the compositions Se-As₂Se₃, As₂Se₃-As₅Se₅, and AsSe-As have eutectic characteristics. Orig. art. has 4 figures.

ASSOCIATION: None

SUBMITTED: 01Jul63

DATE ACQ: 31Mar64

ENCL: 01

SUB CODE: ML, PH

NO REF SOV: 005

OTHER: 000

Card 2/32

ACCESSION NR: AP4036969

s/0078/64/009/005/1174/1181

AUTHOR: Luzimaya, N. P.; Slavnova, G. K.; Medvedeva, Z. S.; Yeliseyev, A. A.

TITIE: The In-As-Se system

SOURCE: Zhurnal neorganicheskoy khimii, v. 9, no. 5, 1964, 1174-1181

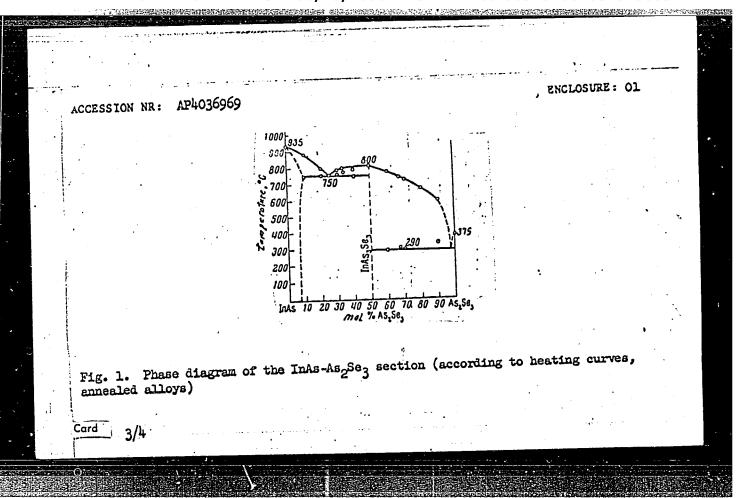
TOPIC TAGS: indium arsenic selenium system, InAs As sub 2 Se sub 3 system, thermal analysis, x ray analysis, microstructural analysis, InAs sub 3 Se sub 3, thermogram, solid solution, InAs, phase diagram, liquidus surface diagram

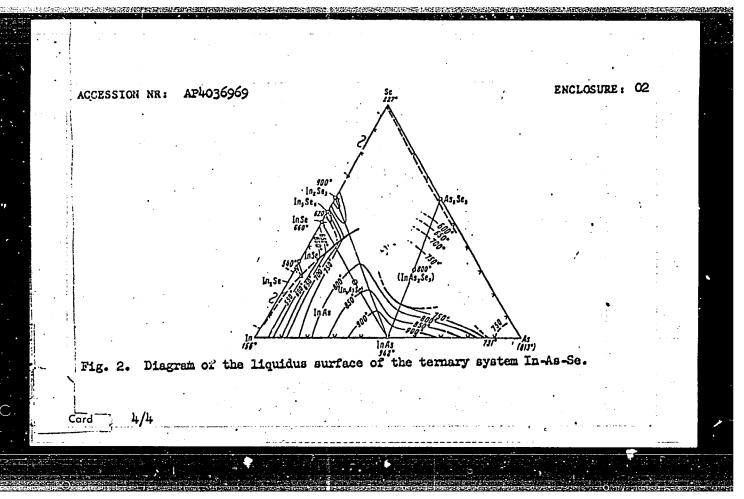
ABSTRACT: The nature of the reactions of the components of the ternary system In-As-Se along the InAs-As₂Se₃ section was studied by thermal, x-ray and microstructural analyses. The previously unknown ternary compound InAs₃Se, melting congruently at 800C (fig. 1) was found. Thermograms for InAs, InAs 10, 50, and 70 mol% As₂Se₃ and As₂Se₃ are given. Microstructural photographs and x-ray. data for these compositions are shown. There was indicated the existence of a relatively small area of solid solutions based on InAs which contained up to about 10 mol% As₂Se₃. An orienting diagram of the liquidus surface of the ternary system In-As-Se was constructed from the authors' and literature data (fig. 2).

Card

1/4

ACCESSION NR: AP4036969
Orig. art. has: 10 figures and 3 tables.
ASSOCIATION: None
SUBMITTED: 03May63
DATE ACQ: 05Jun64
ENCL: 02
SUB CODE: MM
NO REF SOV: 009
OTHER: 004





APPROVED FOR RELEASE: 06/20/2000 CIA-RDP86-00513R001031010003-2"

ACCESSION NR: AP4036975

\$/0078/64/009/005/1302/1303

AUTHOR: Kalitin, V. I.; Luzhnaya, N. P.; Yarembash, Ye. I.; Zinchenko, K. A.

TITLE: Single crystals of praseodymium and neodymium selenides

SOURCE: Zhurnal neorganicheskoy khimii, v. 9, no. 5, 1964, 1302-1303

TOPIC TAGS: single crystal, rare-earth selenide, praseodymium selenide, neodymium selenide, crystal growth, chemical transport reaction

ABSTRACT: Prse, Ndse, and Nd₂Se₃ single crystals have been synthesized by the previously described diffusion method, using a chemical transport reaction with iodine. Optimum conditions for the reactions were established empirically. Habitus of the crystals and x-ray crystallographic data are indicated. The Nd₂Se₃ rhombic crystals were obtained for the first time. Orig. art. has: 1 figure.

Card 1/2

CONTRACTOR OF THE PROPERTY OF

ASSOCIATION: Institut obshchey i neorganicheskoy khimii im. N. S. Kurnakova, Akademii nauk SSSR (Institute of General and Inorganic

Chemistry, Academy of Sciences, SSSR)

SUBMITTED: 04Nov63

ACCESSION NR: AP4036975

DATE ACQ: 05Jun64

ENCL: 00

SUB CODE: 191,85

NO REF SOV: 000

OTHER: 003

Card | 2/2

ADENT PATE	UR/0030/65/000/001/0066/0067
accession não ap5018615	
AUTHORS: Luzhnaya, N. P. (Doctor of che	emical sciences); Koptsik, V. A. (Doctor
of chemical eciances)	59
TITLE: Symposium on crystallization fr	om solutions and melts
SOURCE: AN SSSR. Vestnik, no. 7, 1965,	66-67
TOPIC TACS: crystallization, synthetic confuctor crystal, titanate, tungstate	material, garnet, corundum, mica, semi-
Don't in on March 15-17. One hundred men	on from solutions and melta was held in there from Bulgaria, East Germany, Poland,
the Soviet Union, and Czechoslovakia pa	rticipated, and 24 papers were read. The level questions on theory and method,
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K. Kleber (E. Germany), K. Kh. D. Koppel (SSSR), N. (Czechoslovakia). D. Sch (Gzechoslovakia), and L. done with semidispersed at the Clarger specimens con	Mil (Czechoslovakia). Most eeding or without seeding en ld be obtained with seeding	of the work discussed was
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ENT(m)/ENG(m)/ENP(t)/ENP(b) IJP(c) RDM/JD/JG 8/0363/65/001/001/0053/0056 33546-65 ACCESSION NR: AP5007606 AUTHOR: Kalitin, V. I.; Yarembash, Ye. I.; Luzhnaya, N. P. TITLE: Prasecdymium sesquiselenide SOURCE: AN SSSR. Izvestiya. Neorganicheskiye materialy, v. 1, no. 1, 1965, 53-56 TOPIC TAGS: crystal structure, crystal, rare earth metal, praseodymium, selenide, iodine ABSTRACT: The synthesis (from the elements) and the properties of pure praseodymium selenide, Pr2Se 3 have been studied because indications on the direct synthesis were not found in the literature and only scattered data on the properties and structure of Pr2Se3 are available. Previously praseodymium sesquiselenide was prepared by thermal dissociation of Two methods of synthesis were explored: direct synthesis from pure elements and synthesis by a chemical transport reaction. Direct synthesis was carried out at 800-900°C in an evacuated and sealed quartz ampul containing a powdered mixture of the elements (Pr and Se) in a molar Card 1/3

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ACCESSION NR: AP5007606

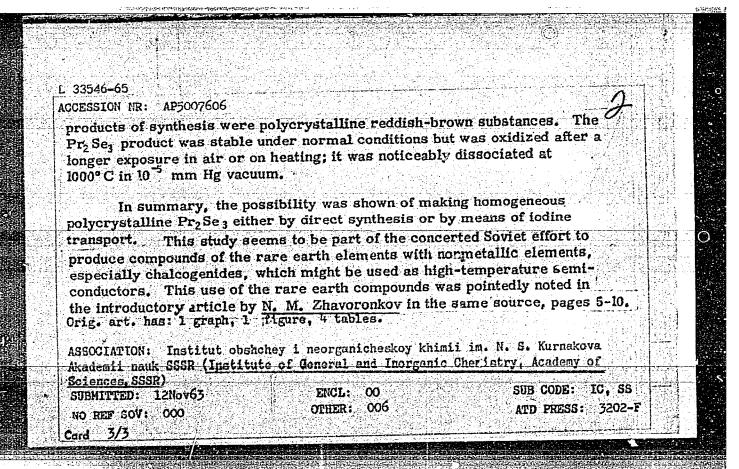
ratio which varied from 3:4 to 1:2. During the process of synthesis, the (heating curves were recorded by differential thermal analysis (DTA). A thermal effect on the curves at 420-435° C indicated an exothermic reaction of explosive type.

The Pr₂Se₃ synthesis by the chemical transport reaction using iodine transport was carried out at a 10°C temperature gradient and T₂ max = 950°C. The synthesized products were iodine free.

Chemical and x-ray phase analysis of the products of both reactions indicated formation of 1) a homogeneous Pr₂Se₃ phase at a stoichiometric Pr:Se ratio or at a ratio corresponding to a slight Pr excess, and 2) a PrSe₂ phase in admixture to Pr₂Se₃ at all other Pr:Se ratios within the range studied.

The x-ray data confirmed the previously reported crystal structure, for Pr_2Se_3 , i. e., a Th_3P_4 -type structure with vacancies in the cationic sublattice. The lattice constant a was determined to be 8.881 kX. The

Card 2/3



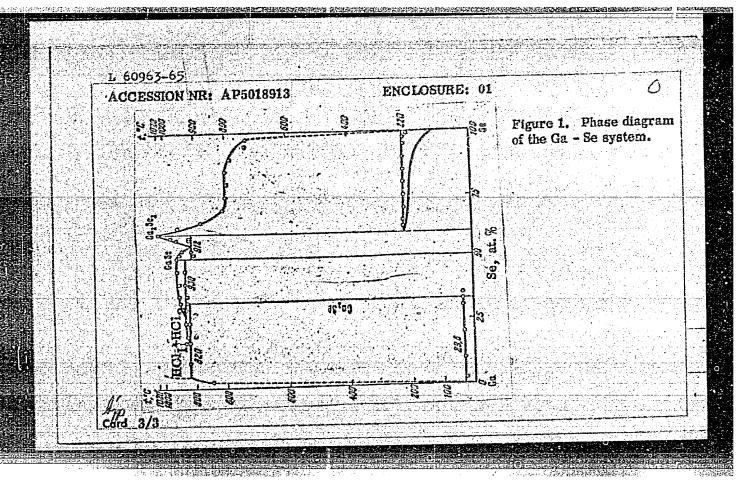
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псшоз	um nk: AP50	/EPA(s)-2/ENT(NN/JG/AT 09366		UR/0	363/65/001/	02/0188/019	
AUTHOR	Gubskaya,	G. F.; Wang, F				A5/0T88\0T8	2 <u>6</u> (
TITLE.			THE-Hall; Luz	nnaya, N. P.	Kudryavtse	v, D. L.	OQ I
		s ru wg-R(III)	-C(V) ternar	V Systems			9
SOURCE:	AN SSSR.	Izvestiva. Ve	Obgani ob a di				
TR8-135		Izvestiya. Ne	A*RdiiTrii68KI	ya materialy,	v. 1, no.	2, 1965,	
of new a AggGaSbz which ta were pro quartz a and micr	semiconductor a AggInAs2 a ke place in duced by mel mpules using	iisgram, phase allium, semico of A(I)-B(III s. The purpos and Ag ₃ GaAs ₂ co alloys with the ting together vibration mix analysis and the ting thermal tr	I)-C(V) type se of this re impounds and e composition the appropri ing. These	systems is o search was to the study of n of these cate elements allows were	f interest in production the chemical products. If in evacuate then contains the co	or production of AggInSt l reactions he alloys d sealed	2 1

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crystallize first from the tables and 5 figures. ASSOCIATION: Institut cbs Akademii nauk SSSR (instit	melt followed by the eute hohey i neorganicheskoy kh ute of General and Inorgan	etic. Orig. art. has: 5
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crystallize first from the tables and 5 figures. ASSOCIATION: Institut cha	melt followed by the eute hohey i neorganicheskoy kh ute of General and Inorgan	imii im. N. S. Kurnakova

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	0R/0383/65/001/006/0843/0622 38
AUTHOR: Rustamov, P. G.; Babaye	eva, B. K.; Luzhnaya, N. P.
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TITLE: Interaction of gallium and s	55°51
SOURCE: AN SSSR. Izvestiya. Neon	rganicheskiye materialy, v. 1, no. 6, 1965, 843-844
TOPIC TAGS: gullium, selenium, g	한다회에 선생님의 전환 시간 아이들에 가장 사람들이 되지 않는데 가는 사람들이 되었다. 그는 그 사람들이 되었다.
ABSTRACT: The Ga - Se system w	as studied over a wide concentration range. The printal and x-ray phase analysis; in addition, the micro-
given in Fig. 1 of the Enclosure, w	as protect on the basis of the Phase separation
occurs between 4 and 17 at. % Se at	920C. The compound GazSe is formed via a peritect
reaction between the melt, which co	Jacks and A subjecte is formed between Gase
	. On the Ga_2Se_3 - Se section of the diagram (60-100 e and has a melting point of 220 \pm 5C. The thermal
	a and had a mairing noine of 240 I due the there

ACCESSION NR: AP5018913 analysis data are in complete ag The powder patterns obtained for The difference in the powder pat confirms the individuality of this ASSOCIATION: Institut khimii A	tern of Ga ₂ Se from the patter s compound. Orig. art. has:	rns of Ga ₂ Se ₃ and GaSe 1 figure.
of Sciences AzerbSSR) 65 SUBMITTED: 26Feb65	encl: 01 Other: 003	SUB CODE: IC,
NO REF SOV: 004		

"APPROVED FOR RELEASE: 06/20/2000 CIA-RDP86-00513R001031010003-2



L 15207-66 EMT(m)/T/EMP(t)/EMP(b) LJP(c) JD/JG

ACC NR: AP6001299 SOURCE CODE: UR/0363/65/001/008/1328/1334

AUTHOR: Luzhnaya, N. P.; Nikol'skaya, G. F.; Wang, Ping-nan

ORG: Institute of General and Inorganic Chemistry im. N. S. Kurnakov, Academy of Sciences SSSR (Institut obshchey i neorganicheskoy khimii Akademii nauk SSSR)

TITLE: Semiconducting compounds of type A B HIC V

SOURCE: AN SSSR. Izvestiya. Neorganicheskiye materialy, v. 1, no. 8, 1965, 1328-1334

TOPIC TAGS: copper compound, gold compound, arsenic compound, indium compound, gallium compound, antimony compound

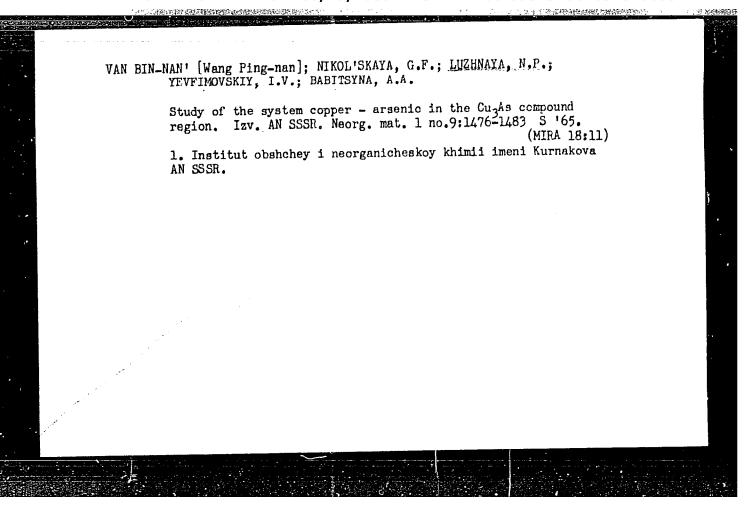
ABSTRACT: An attempt was made to prepare the compounds Cu3BHC2V and Au3InC2V, where BH is indium or gallium, and CV is arsenic or antimony, by fusing together the elements taken in stoichiometric proportions in evacuated ampoules with vibratory strying. Phase diagrams of binary systems entering into the ternary system copper-gallium-arsenic were studied. To determine the interaction in alloys of the composition Cu3GaAs2, the section GaAs-Cu; A= 3: 1 of the Cu-Ga-As system was investigated, since, based on the phase diagrams of the binary systems, Cu3GaAs2 should lie on this section. Alloys corresponding to the compositions Cu3GaAs2, Cu3GaAs2, Cu3GaSb2, and Cu3InSb2 did not consist of a single phase) N. e.; ternary compounds of these compositions are not formed under the conditions studied. Thermographic and microstructural data also indicate that alloys of the

Card 1/2

criteria for	ns Au ₃ InAs ₂ and a predicting the er oup of the first gr	roup of the perio	dic system in the	le phase eithounds should e case of con	er. Apparent be confined to apounds of type	iy, the o the
A3BIII C2V	Orig. art. has:	7 figures and 2	tables.			
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1 12						

EWA(k)/FBD/EWT(1)/EWT(m)/EEC(k)-2/T/EWP(t)/EWP(k)/EWP(b)/EWA(m)-2/EWA(h) L 3976-66 UR/0363/65/001/009/1484/1492 SCIB/IJP(c) WG/JD/JG ACC NRIAP5025781 546.27'181.1 AUTHOR: Grinberg, Ya. Kh., Zhukev, E. Luzhnaya, Medvedeva, S. : TITLE: Kinetics of the reaction of amorphous boron with phosphorus SOURCE: AN SSSR. Izvestiya. Neorganicheskiye materialy, v. 1, no. 9, 1965, 1484-1492 maser sem semiconductor, boron phosphide, boron compound, kinetics, TOPIC TAGS: rectifier reaction mechanism ABSTRACT: Boron phosphide (BP) is of considerable interest since rectifiers made from it can function in an oxidizing atmosphere at up to 1000C. Boron phosphide monocrystals may prove useful for the design of masers and similar devices. In this work, the reaction of boron with phosphorus vapor was studied at 1000, 1100, and 1150C. It was found that the reaction is initially rate controlled and follows second-order kinetics. Following a transition period, the reaction becomes diffusion controlled and obeys first-order kinetics. The latter stage of the reaction is presumably caused by the formation of a coating on the boron. The rate constants and activation energies of both reaction stages were determined. A mechanism is proposed for the reaction. The optimum quality of BP (< 10-3% Si) was obtained when the reaction was conducted at 1150-12000 for 1 hr or less, using amorphous boron. Orig. art. has: 7 figures, 3 tables, and 10 formulas. ASSOCIATION: Institut obshchey i neorganicheskoy khimii im. N. S. Kurnakova Akademii nauk SSSR (Institute of General and Inorganic Chemistry, Academy of Sciences, SSSR)

ACC NR.475025781		
SUBMITTED: 29Apr65	encu: 00	SUB CODE: 55,6C
NO REF SOV: 007	OTHER: 012	ATD PRESS: 4//8
	도 하시아 있는 것들이 한 경험을 받는 것도 없다. 한 1일 시간 중에 이 전로 발생한 것 같은 것이다.	당도 시작을 살이 그 그 그리는 것 같습니다. (설립 및 100호 - 프로그램의 스토트)
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Card 2/2		



GRINBERG, Ya.Kh.; ZHUKOV, E.G.; MEDVEDEVA, Z.S.; LUZHNAYA, N.P.

Kinetics of interaction of amorphous boron with phosphorus. Izv. AN SSSR. Neorg. mat. 1 no.9:1484-1492 S *65. (MIRA 18:11)

1. Institut obshchey i neorganicheskoy khimii imeni Kurnakova AN SSSR.

1. 16715-66 DWT(m)/EWP(t) LJP(c) JD SOURCE CODE: UR/0078/65/010/010/2315/2319

AUTHOR: Koppel, Kh. D.; Luzhnaya, N. P.; Medvedeva, Z. S.

35 B

ORG: none

TITLE: The Cd-In-As system

SOURCE: Zhurnal neorganicheskoy khimii, v. 10, no. 10, 1965, 2315-2319

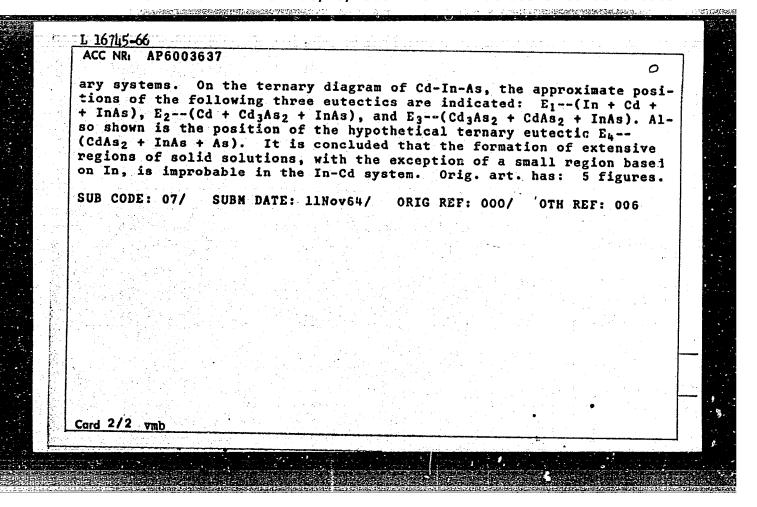
TOPIC TAGS: cadmium, indium, arsenic, phase diagram, indium compound, arsenic compound, cadmium compound, arsenide

ABSTRACT: Some sections of the Cd-In-As system were studied by differential thermal and microstructural methods. In the thermal analysis, use was made of InAs, Cd₃As₂, CdAs₂, and cadmium metal. The sections InAs-Cd, InAs-Cd₃As₂, InAs--(50 at % Cd + 50 at % As), and InAs-CdAs₂ were studied. Phase diagrams were used to plot the diagram of the surface of the liquidus of the Cd-In-As system. As in the case of the Zn-

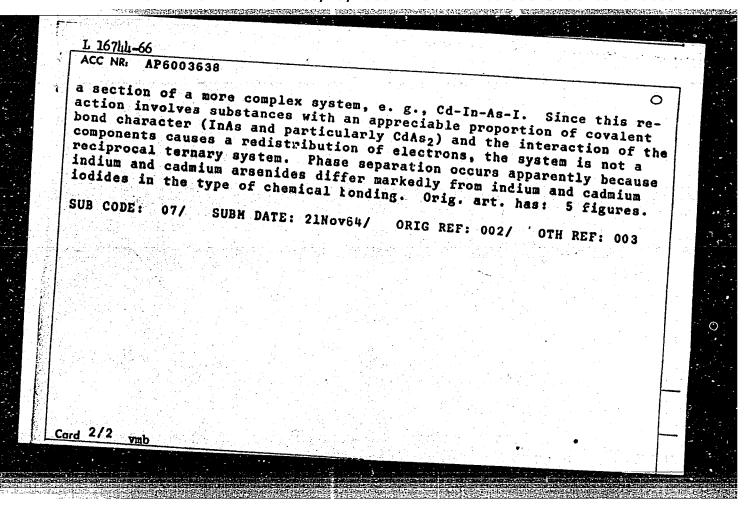
-Ga-As system, the field of primary crystallization of the ${\rm A^{III}_{B}}^{V}$ compound, in this case indium arsenide, occupies the major portion of the diagram. The comparatively small fields of primary crystallization of In, Cd, Cd₃As₂, CdAs₂, and As are located next to the corresponding bin-

UDC: 541.123+546.48+546.682+546.19

Card 1/2



IJP(c) JD_ FMT(m)/EWP(t) SOURCE CODE: UR/0073/65/010/010/2320/2323 I. 167111-66 AP6003638 AUTHOR: Luzhnaya, N. P.; Hedvedeva, Z. S.; Koppel, Kh. D. ORG: none TITLE: Reaction of indium arsenide with cadmium iodide 27 Zhurnal neorganicheskoy khimii, v. 10, no. 10, 1865, 2320-2323 TOPIC TAGS: cadmium compound, arsenide, iodide, indium compound ABSTRACT: The reaction between InAs and CdI2 was studied at 1000°C. Thermal analysis of 18 different compositions of InAs-CdI2 mixtures was performed and the data were used to plot a phase diagram of the InAs--CdI2 system. Microstructural analysis showed the presence of two layers, with different properties, caused by phase separation in the liquid state. It also showed the presence of two phases in each layer. X-ray and chemical analysis showed that the lower layer consisted of InAs and CdAs2 and the upper layer of CdI2 and InI. The data indicated that the reaction of InAs with CdI2 occurs in accordance with the reaction 2InAs + CdI₂ = 2InI + CdAs₂ whose equilibrium shifts neither left nor right. This system should not be seen as a binary system but rather as UDC: 546.682'19 + 546.48'151 Card 1/2



(A) L 27859-66 EWT(1)/EWT(m)/T/EWP(t)/EWP(b)/EWA(c) IJP(c) JD/JG/GG AP5028625 ACC NR SOURCE CODE: UR/0030/65/000/010/0049/0054 AUTHOR: Luzhnaya, N. P. (Doctor of chemical sciences); Yarembash, Ye. I. (Candidate of chemical sciences); Fedvedeva, Z. S. (Candidate of chemical sciences) ORG: Institute of General and Inorganic Chemistry im. N. S. Kurnakov, Academy of Sciences, SSSR (Institut obshchey i neorganicheskoy khimii Akademii nauk SSSR) TITLE: Method of transport reactions in semiconductor chemistry SOURCE: AN SSSR. Vestnik, no. 10, 1965, 49-54 21,44155 TOPIC TAGS: single crystal growing semiconductor single crystal, semiconducting film, boron compound, phosphide, selenide, telluride, rare earth element, semiconducting material, refractory, single crystal, chemical reaction ABSTRACT: Since 1962, the semiconductor chemistry laboratory of the Institute of General and Inorganic Chemistry im. N. S. Kurnakov, Academy of Sciences SSSR (laboratoriya khimii poluprovodnikov Instituta obshchey i neorganicheskoy khimii Akademii nauk SSSR) has been conducting systematic research on growing single crystals of boron phosphide and rare earth selenides and tellurides by the method of transport reactions. The mechanism of these reactions is explained, and a description of the preparation of boron phosphide (BP) in the form of single crystals and polycrystalline layers is given. Also discussed is the preparation of chalcogenides of elements of the cerium group having the composition Me2X3 and MeX2 and characterized by semiconducting properties. It is concluded that the method of transport reactions for growing single crystals and films of refractory semiconductors has great UDC: 621.315.52

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ACC NR: AP6032946

SOURCE CODE: UR/0363/66/002/010/1747/1756

AUTHOR: Zinchenko, K. A.; Luzhnaya, N. P.; Yarembash, Ye. I.; Yellseyev, A. A.

ORG: Institute of General and Inorganic Chemistry im. N. S. Kurnakov, Academy of Sciences, SSSR (Institut obsnchey i neorganicheskoy khimii Akademii nauk SSSR)

TITLE: Phase diagram and phase properties of the Nd-Te system

SOURCE: AN SSSR. Izvestiya. Neorganicheskiye materialy, v. 2, no. 10, 1966, 1747-1756

TOPIC TAGS: neodymium compound, telluride, semiconductor single crystal, polycrystal, single crystal structure, neodymium tellurida neutronductor, phase diagram, place composition, mutai projectly, electric neithered, crystal lettice defect
ABSTRACT: The phase composition and physical properties of Nd—Te alloys have been studied over the entire range of compositions. The stated purpose of the study was to refine the previously established phase diagram of the Nd—Te System [Ye. I. Yarembash, A. A. Yeliseyev, K. A. Zinchenko, Zh. neorgan. materialy, v. 1, no. 1, 1965, 60 and N. Kn. Abrikosov, V. Sh. Zargaryan. Zh. neorgan. materialy, v. 1, no. 9, 1965, 1462] and to determine the phase-composition dependence of electrophysical properties of the polycrystalline alloys and of certain single crystals. The complete phase diagram of the Nd—Te System, which was plotted on the basis of new experimental data, was basically similar to that previously established by the authors. The existence of seven individual phases, isostructural with the corresponding La phases,

Card 1/2

UDC: 541.123.2

ACC NR: AP6032946

was confirmed. New crystallochemical x-ray data were determined for Nd₁Te₇ and NdTe₃ phases: A polymorphic transition was detected by x-ray in the Nd₂Te₃ samples in contrast with the M₂Te₃ compounds of the ceria group elements which precede Nd in the Periodic Table. Melting points of certian phases differ significantly with the earlier Soviet data. Electrical resistivity of the phases in the Nd—Te System continuously increased with an increase in the Te content of the samples. Semiconductor property and n-type conductivity were confirmed in all neodymium tellurides. Carrier concentration varied from 10^{21} cm⁻³ for NdTe to 10^{18} cm⁻³ for NdTe₃. A defective lattice in Nd₂Te₃ and Nd₄Te₇ was confirmed by the resistivity, thermal conductivity, and most of all, by the coefficient of thermal emf data. Single crystals of Nd₃Te₄, Nd₄Te₇, NdTe₂, and NdTe₃ were grown to obtain purified samples for determining semi-conductor characteristics. Orig. art. has: 4 figures and 5 tables.

SUB CODE://20/ SUBM DATE: 09Dec65/ ORIG REF: 007/ OTH REF: 005/

Card 2/2

ACC NRI AP6036783

SOURCE CODE: UR/0363/66/002/011/1930/1938

AUTHOR: Kalitin, V. I.; Yarembash, Ye. I.; Luzhnaya, N. P.

ORG: Institute for General and Inorganic Chemistry im. N. S. Kurnakov, AN SSSR (Institut obshchey i neorganicheskoy khimii AN SSSR)

TITIE: Phase diagram of the praseodymium-selenium system

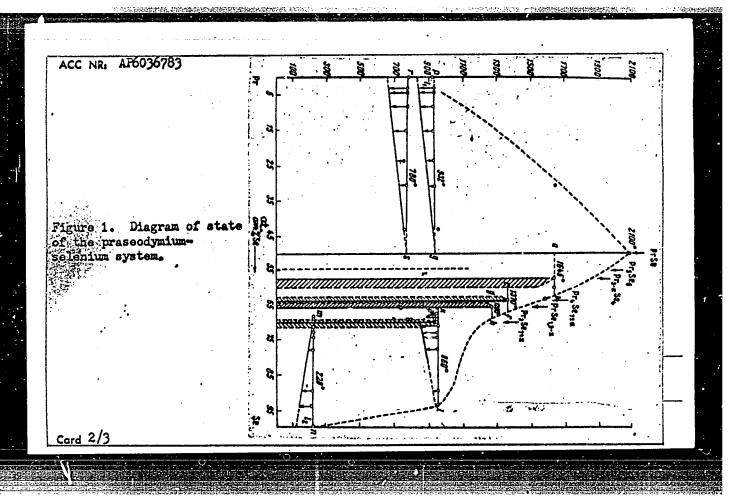
SOURCE: AN SSSR. Izvestiya. Neorganicheskiye materialy, v. 2, no. 11, 1966, 1930-1938

TOPIC TAGS: praseodymium, selenium, alloy phase diagram

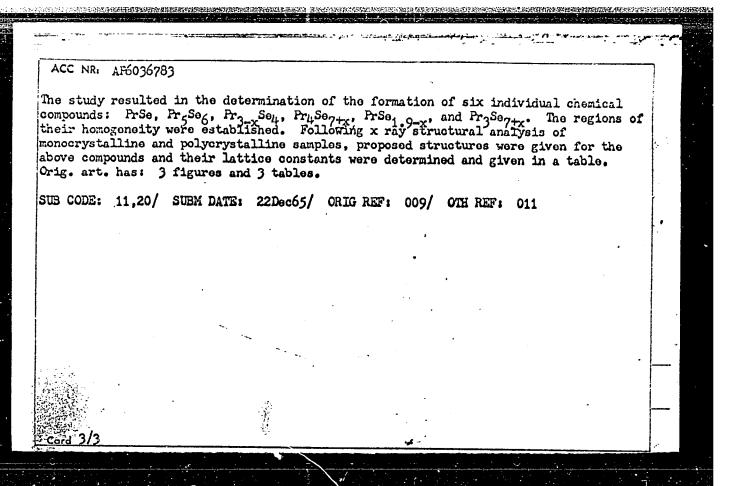
ABSTRACT: The article starts with a table, based on literature data, giving the properties of the known praseodymium selenides. An experimental investigation was made of powder form and fused preparations of praseodymium selenides, the composition of which varied from 0 to 100 at. \$\mathcal{F}\$ selenium, as well as of single crystals obtained by various methods. For determination of the phase diagram, thermal, x ray, and microstructural analysis was used. The detailed results are given in tabular form and are best summarized by the diagram shown. (See Fig. 1)

Card 1/3

UDC: 546.656+546.23



APPROVED FOR RELEASE: 06/20/2000 CIA-RDP86-00513R001031010003-2"



ACC NR: AP7002399

SOURCE CODE: UR/0363/66/002/012/2130/2133

AUTHOR: Grinberg, Ya. Kh.; Luzhnaya, N. P.; Medvedeva, Z. S.

ORG: Institute of General and Inorganic Chemistry im. N. S. Kurnakov, Academy of Sciences, SSSR (Institut obshchey i neorganicheskoy khimii Akademii nauk SSSR)

TITLE: Study of the equilibrium in the boron phosphide - iodine system

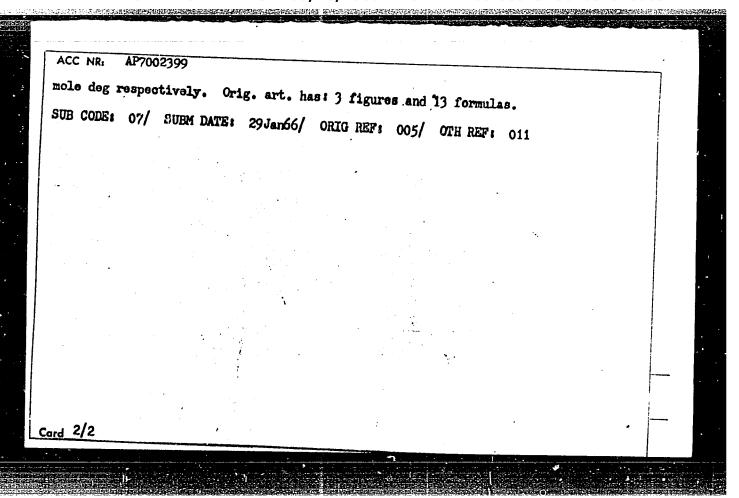
SOURCE: AN SSSR. Izvestiya. Neorganicheskiye materialy, v. 2, no. 12, 1966, 2130-

TOPIC TAGS: boron compound, phosphide, iodine, chemical equilibrium

ABSTRACT: The heterogeneous equilbirium between solid boron phosphide and gaseous iodine was studied at 1075, 1120, 1160 and 1195°C. The amount of phosphorus and boron in the gas phase were determined from the weight loss of the solid phase, and the amount of iodine introduced was known. Assuming the equilibrium reaction to be 2BPs + BI3g = 3BIg + P2g, the authors calculated the equilibrium constant K_p of this reaction, $K_p = [p_{BI}]^3$ pp./pgr. Within the limits of experimental error, K_p thus calculated for all four temperatures had the same value, and its temperature dependence is range studied log K_p varies linearly with reciprocal temperature. The enthalpy ΔH and entropy ΔS of the reaction per mole of BP were found to be 44 kcal/mole and 24 cal/

Card 1/2

UDC: 546.27 181.1+546.15



AUTHOR: Lummaya, N. F.

A-piloufa;

OliGa none

TITLE: Results of testing different numerical schemes for predicting the pressure field

SOURCE: Noscow. Reentralinyy institute prognozov. Trucy, no. 130, 1966. Sinopticheskaya meteorologiya (Synoptic meteorology), 46-53

TORIC TAGS: westher forecasting, atmospheric pressure, atmospheric model.

ABSTRACT: A statistical and symoptic evaluation of forecast charts prepared by the Kurbatkin, Kadyshnikov, and Bortnikov models is given. In the Kadyshnikov and Bortnikov models the pressure field is predicted by means of full hydrodynamic and thermodynamic equations. For this reason, these schemes are most promising, since they are based on more precise physical models of the atmosphere. The Kadyshnikov model predicts the field at the earth's surface and at the 500- and 300-militoar levels, the Bortnikov model at the 700- and 300-militar levels. The Kurpatkin scheme is a five-level quasi-geostropic model, taking into account the scratification of the air. It is important because it predicts for a large number of levels (earth's surface and the 650-, 700-, 500-, and 300 millibar levels). Comparisons

Card 1/2

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of results, using all three models, show that the Boronikov model gives best results at the 700-millibar level. All give similar results at the 500- and 300-millibar levels, but the pressure centers are somewhat better defined at 500 millipara on the Kadyshnikov model. However, the latter also gives smoother high-pressure fields than the others, and thus fails to reflect actual cyclones and ambicyclones. The author believes that refinements in the Madyshrikov and Bortnikov models will. make them the most promising for predicting the presonce field. Orig. art. heat 7 vables.

SUB CODE: OL/ SUBN DATE: none/

9075 HB; 607

ACC NR: AP7008524

SOURCE CODE: UR/0363/67/003/002/0300/0310

AUTHOR: Koppel, Kh. D.; Modvodova, Z. S.; Luzhnaya, N. P.

ORG: Institute of General and Inorganic Chemistry im. N. S. Kurnakov, Academy of Sciences, SSSR (Institut obshchey i neorganicheskoy khimii Akademii nauk SSSR)

TITLE: Reaction of indium arsonide with cortain metals

SOURCE: AN SSSR. Izvestiya. Neorganicheskiye materialy, v. 3, no. 2, 1967, 300-310

TOPIC TAGS: indium compound, arsenide, phase diagram, solubility

ABSTRACT: The liquidus surfaces of the ternary systems Zn-In-As, In-Sn-As and In-Pb-As were plotted in order to find solvents for the crystallization of indium arsenide. The system Cd-In-As, studied earlier, is also considered. The criteria for determining the suitability of these systems for the crystallization of InAs were: 1) type of diagram, 2) adequate solubility of InAs in the liquid solvent, 3) minimum content of InAs in the eutectic composition and 4) possibility of separating InAs crystals from the solvent. It was found that InAs is substantially soluble in the liquid state at relatively low temperatures in a series of sections of the systems studied. The choice of crystallization conditions is determined both by the lower liquidus temperature on the section and by the fact that the InAs crystals can be completely separated from the solvent. The crystallization conditions are more favorable if the third component of the system melts at relatively low temperatures, and InAs deter-

Card 1/2

UDC: 546.682 191+546.3

APPROVED FOR RELEASE: 06/20/2000

CIA-RDP86-00513R001031010003-2"

ACC NR: AP7008524

mines the triangulation of the system. The study of the ternary systems made it

possible to grow InAs crystals from several solvents by spontaneous growth, Bridgman's method, and the temperature gradient method. In spontaneous crystallization, the best solvents were found to be the compositions InAs-In-M (M = Cd, Sn, Po). Orig. art. has: 11 figures.

SUB CODE: 07/ SUBM DATE: 06Jun66/ ORIG REF: 013/ OTH REF: 019

Card 2/2

CERASINENKO, T.N., dotsent; INZENIAYA, R.M., doktor.

Gordeev's solution in treating cancer of the eyelids and conjunctive. Vest. oft. 68 no.1:25-27 Js-F '56. (HURA 9:5)

1. Iz glaznoy klinik Stanislavskogo meditsinskogo instituta.

(ANTISEPTICS

Gerdeev's solution, ther. of cancer of eyelids & of conjunctiva)

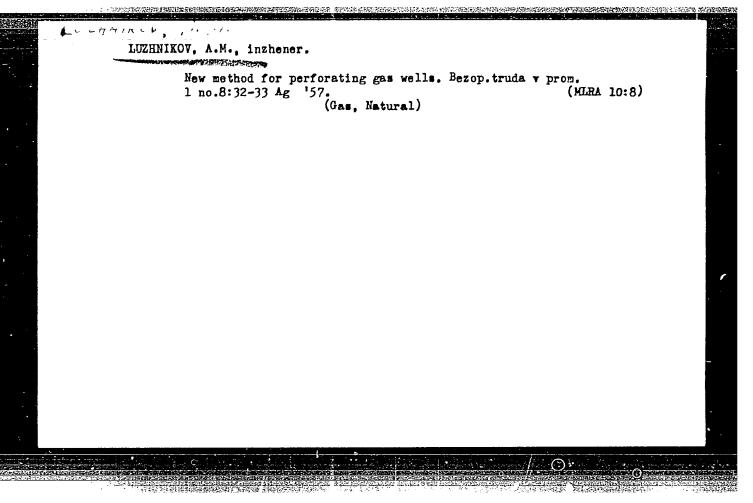
(EYELIDS, neoplasms ther., Gordeev's solution)

(CONJUNCTIVA, neoplasms same)

LUZHNAYA, R.M.

Use of Gordeev's solution No.1 in corneal ulcers. Oft.zhur.
13 no.3:153-158 *58 (MIRA 11:6)

1. Iz kliniki glaznykh bolezney (zav. - dots. T.V. Shlopak) Stanislavskogo meditsinskogo instituta. (CORNEA--ULCERS)



KUTUKOV, A.I., red.; ZAYTSEV, A.P., red.; DROGALIN, G.V., red.; POLESIN, Ya.L., red.; KOSTYUKOV, N.N., red.; KURAS, D.M., red.; LUZHNIKOV.

A.M., red.; RODIONOV, I.S., red.; BLOKH, S.S., red.; SULTANOV, D.K., red.; BIBILUROV, V.P., red.; PETROV, A.I., red.; KHARCHEVNIKOV, N.M., red.; ANDRIANOV, K.I., red.; GADZHINSKAYA, M., red.; zd-ya: BERESLAVSKAYA, L.Sh., tekhn.red.

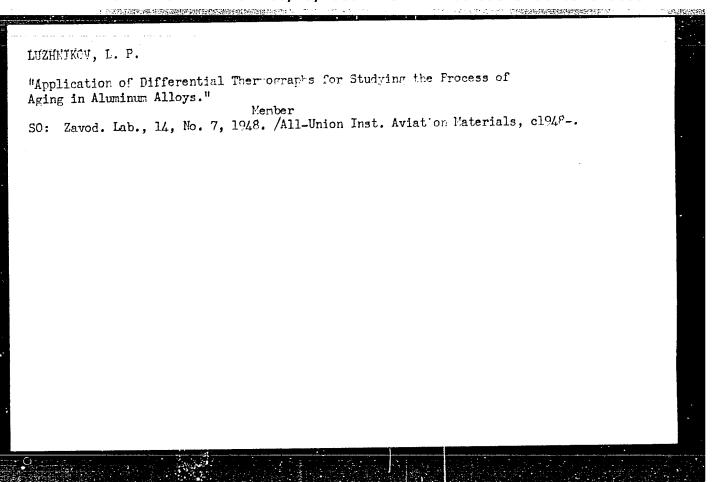
[Safety regulations for petroleum and gas producing industries]
Pravila bezopasnosti v neftegazodobyvaiushchei promyshlennosti.
Moskva, Gos.nauchno-tekhn.izd-vo lit-ry po gornomi delu, 1960.
123 p. (MIRA 14:3)

1. Russia (1917- R.S.F.S.R.) Gosudarstvennyy komitet po nadzoru za bezopasnym vedeniem rabot v promyshlennosti i gornomu nadzoru.

2. Tšentral'nyy apparat Gosgortekhnadzora RšFšR (for Kutukov, Zaytsev, Drogalin, Polesin, Kostyukov, Kuras, Luzhnikov, Rodionov, Blokh). 3. Vsesovuznyy nauchno-issledovatel'skiy institut po tekhnike bezopasnosti (for Sultanov). 4. Upravleniya ukrugov Gosgortekhnadzora RšFšR (for Bibilurov, Petrov, Kharchevnikov).

5. Tšentral'nyy komitet profsoyuza rabochikh neftyanoy i khimicheskoy promyshlennosti (for Andrianov).

(Oil fields--Safety measures)



EVERHART, John L.; GLAZUNOV, S.G., [translator], redaktor; LUZENIKOV,
LAR., [translator], redaktor; ARKHANGEL'SKAYA, M.S., redaktor;
EVENSON, I.M., tekhnicheskiy redaktor

[Titanium and titanium allys. Translated from the English]
Titan i ego splavy. Perevod s angliiskogo. Moskva, Gos.nauchnotekhn.izd-vo lit-ry po chernoi i tsvetnoi metallurgii, 1956.

138 p. (Titanium)

(MIRA 9:3)

AL'TMAN, Morits Borisovich; LEREDEV, Aleksandr Aleksandrovich; POLYANSKIY,
Aleksey Pavlovich; GHUKHROV, Matvey Vasil'yevich; HIKHEYEVA, V.I.,
professor, doktor, retsenzent; KRYMOV, V.V., kamidat tekhnicheskikh
nauk, retsenzent; FRIDLYANDKR, I.N., kandidat tekhnicheskikh nauk,
retsenzent; TELIS, M.Ya, inzhener, retsenzent; KRYSIN, B.T., retsenzent;
LUZHNIKOV, L.P., redaktor; KAMAYEVA, O.M., redaktor izdatel'stva;
ATTOPOVICH, M.K., tekhnicheskiy redaktor

[Melting and casting of light alloys] Plavka i lit'e legkikh splavov.

Moskva, Gos. nauchno-tekhn. izd-vo lit-ry po chernoi i tavetnoi
metallurgii, 1956. 491 p.

(Alloys--Metallurgy)

LUZHNIKOV, L. P. and ROMANOVA, O. A.

"New Data on the Role of Manganese in the "Extrusion Effect" in Aluminum Alloys"

Light Alloys. no. 1: Physical Metallurgy, Heat Treatment, Casting, and Forming; Principal Reports of the Conference, Moscow, Izd-vo AN SSSR, 1958, 497 P.

(2nd. A.U. Conf on Light alloys ASS)

SOV/137-58-9-20038

Translation from: Referativnyy zhurnal, Metallurgiya, 1958, Nr 9, p 277 (USSR)

AUTHORS: Luzhnikov, L.P., Romanova, O.A.

TITLE: New Data on the Role of Manganese in Connection with the

Press Effect in Aluminum Alloys (Novyye dannyye o roli margantsa v svyazi s presseffektom v alyuminiyevykh splavakh)

PERIODICAL: V sb.: Legkiye splavy. Nr l. Moscow, 1958, pp 245-248

ABSTRACT: Doubt is cast on the hypothesis that the press effect (P), all

other conditions being equal, can be observed only in alloys (A) containing Mn or some other element tending significantly to increase the recrystallization temperature of Al A. A number of A are investigated, including Al-Cu-Mn A, over a fairly broad range of Cu and Mn contents. These A not only failed to display the P, but, on the contrary, the properties of cold-formed semisinished products treated under optimum conditions of artificial aging are significantly higher than those of extruded items. When tenths of one per cent of Mg are added to Al-Cu-Mn alloys, normal P is observed. Metallographic

investigation confirms the existence of differences in the Card 1/2 recrystallization process of Al-Cu-Mn alloys with and without

SOV/137-58-9-20038

New Data on the Role of Manganese (cont.)

Mn. Results characteristic of other A are obtained in investigation of A having the following % contents: Cu 4.8, Mn 0.85, Ti 0.1, and 0.20-0.25 each of Fe and Si. The need for a more penetrating study of the effect of phase composition on the P of Al A is emphasized.

G.T.

1. Aluminum alloys---Properties 2. Manganese---Metallurgical effects 3. Aluminum alloys---Theory

Card 2/2

LUZHNIKOV, L.P.

PHASE I BOOK EXPLOITATION

SOV/3505

Spravochnik po mashinostroitel'nym materialam v chetyrekh tomakh, tom 2: Tsvetnyye metally i ikh splavy (Handbook on Machine-Building Materials in 4 volumes, v. 2: Nonferrous Metals and Alloys) Moscow, Mashgiz, 1959. 639 p. Errata slip inserted. 25,000 copies printed.

Ed.: G. I. Pogodin-Alekseyev, Doctor of Technical Sciences, Professor; Ed. of this vol.: M. A. Bochvar, Engineer; Ed. of Publishing House: V. I. Rybakova, Engineer; Managing Ed. for Information Literature: I. M. Monastyrskiy, Engineer; Tech. Eds.: T. F. Sokolova and B. I. Model'.

FURPOSE: This book is intended for machine designers and metallurgists.

COVERAGE: The book presents comprehensive tabular and textual data on the chemical composition, physical and mechanical properties, microstructure, heat treatment, applications, etc., of verious nonferrous metals and alloys used in machinery manufacture. Metals dealt with are aluminum, magnesium, copper, nickel, cobalt, titanium, zinc, and cadmium, together with certain precious and rare metals. Special materials considered are hard alloys (including sintered carbides), cermets, and ply metals. Special alloys, such as bearing, Card 1/22

andbook on Machine-Building (Cont.)	OF/3505
casting, corrosion-resistant, heat-resistant, electrical resistant fusible alloys, as well as solders, are treated. Authors articles are listed in the table of contents. Various refere both Soviet and non-Soviet, are scattered throughout the book	of neas,
ABLE OF CONTENTS:	
h. I. Aluminum and Its Alloys	7
Aluminum (Luzhnikov, L. P., Candidate of Technical Sciences) Workable aluminum alloys	7 9
Alloys included in GOST standards	9
Alloys AD and ADL Alloy AMts	12 20
Alloy AMg Alloy AMg5p Alloy AMg3	21. 22 22
Alloy AMg5	23
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Landbook on Machine-Building (Cont.)	SOV/35 05
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Alloy AMg7	24
Alloys D1 and D1P	24
Alloy D6	27
Alloy D16 and D16P	27
Alloy D3P	31
Alloy DI&P	37
Alloy AV Alloy AK	3L
Alloy AK2	24 2h
Alloy AK4	27 31 31 34 34 35 36 37 39 40
Alloy AK4-1	36
Alloy AK6	37
Alloy AK8	39
Alloy V95	40
Alloys not included in GOST standards	43
Alloy V65	43
Alloy VD17	44
Alloy D20	45
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SOV/129-59-3-2/16

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Luzhnikov, L.P., Condidate of Technical Sciences AUTHORS:

and Novikova, V.M., Engineer

Relations Governing the Changes in the Mechanical and TITLE:

the Technological Properties of Ternary Titanium Base Alloys (Zakonomernosti izmeneniya mekhanicheskikh i tekhnologicheskikh svoystv troynykh splavov na osnove

titana)

Metallovedeniye i Termicheskaya Obrabotka Metallov, PERIODICAL:

1959, Nr 3, pp 6 - 13 (USSR)

The aim of the work described in this paper was to ABSTRACT:

determine the most rational combination of elements in alloying titanium for the purpose of obtaining heatresistant sheet alloys with good welding properties. The system Ti-Al was chosen as the basic one; as the third element, Cr, Mn, Mo, Fe were introduced. In each of these ternary systems, one cut of the diagram of state was studied, which corresponded to a 6% total content of

alloying eléments. For comparison, alloys of the

Ti-Fe-Mn system with a total Fe + Mn content of 6% were

also studied. All the alloys were produced from sponge titanium of a single batch. Sheets produced from the

Cardl/3

SOV/129-59-3-2/16

Relations Governing the Changes in the Mechanical and the Technological Properties of Ternary Titanium Base Alloys

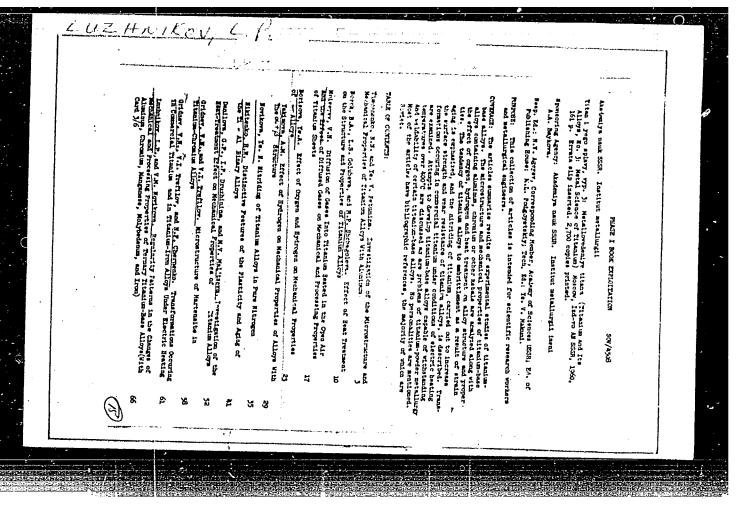
titanium sponge without alloying had a strength of 53.5 kg/mm² and elongation of 26.8% and a contraction of 34.2%. The alloys were studied in the shape of 1 mm thick sheets produced under laboratory conditions. ingots were produced by smelting twice in a neutral atmosphere in arc furnaces with expendable electrodes. The electrodes for the second smelting were forged from the ingcts produced from the first smelting. Then followed forging, hot and "warm" rolling, etching and annealing of the sheets. The finished specimens were then annealed in vacuum for 2 hours at 800 °C, cooled in the furnace to 200 °C and then cooled down in air from that temperature onwards. The mechanical properties of the alloys were determined at room temperature and at 250, 300 and 350 °C. The stamping properties at 20 and 500 °C, as well as the weldability and the properties of the welded joints were also determined. The chemical compositions of the sheets after etching and annealing Card2/3 are entered in Table 1 for 20 heats. The results are

sov/129-59-3-2/16

Relations Governing the Changes in the Mechanical and the Technological Properties of Ternary Titanium Base Alloys

described of the mechanical tests (Figure 1) of the technological properties (bending of a 15 x 100 mm specimen around a radius equal to the sheet thickness until the first crack appears) and the behaviour during stamping (Figure 2) and also of weldability tests (Figure 3). On the basis of systematic study of the mechanical and technological properties and of the weldability of the five ternary titanium base alloys: Ti-Al-Mo, Ti-Al-Cr, Ti-Al-Mn, Ti-Al-Fe and Ti-Fe-Mn, the authors have shown that it is advisable to alloy titanium-aluminium alloys with β stabilisers within limits approaching their Such alloys possess a maximum sclubility in α -titanium. gcod technohigh strength at elevated temperatures; logical plasticity and satisfactory weldability. There are 3 figures, 2 tables and 5 references, 3 of which are English, 1 Soviet and 1 Soviet translation of an English book.

Card 3/3



"APPROVED FOR RELEASE: 06/20/2000

CIA-RDP86-00513R001031010003-2

S/762/61/000/000/002/029

AUTHORS: Luzhnikov, L.P., Novikova, V.M.

Binary titanium-tin and titanium-zirconium alloys. TITLE:

Titan v promyshlennosti; sbornik statey. Ed. by S.G.Glazunov. SOURCE:

Moscow, 1961, 31-40.

The paper reports an experimental investigation, performed in 1957-58, of Ti-Sn and Ti-Zr binary alloys with up to 10% Sn and 8% Zr. Both alloys exhibit a broad range of solid solutions (SS) with a Ti; while both alloys lack practical interest per se, their characteristics are of importance in the understanding of more complex Sn and especially Zr alloys. Mechanical and formability properties (including weldability) were tested on sheet material made from a single batch of sponge Ti, which had a tensile strength of 55 kg/mm² and 32.7% elongation. Details of the preparation and composition of the test alloys are described and tabulated. The 1.3-1.5-mm thick sheets were sand-blasted and etched in a solution of 650 cm³ H₂O, 350 cm³ HCl, and 50 g NaF, at 50-60°C. The test specimens for tensile, bending, stampability, and weldability tests were vacuum-annealed (at 5.10-3 mm Hg, 800°C, 2 hrs), furnace-cooled to 200°C, and then air-cooled.

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CIA-RDP86-00513R001031010003-2" **APPROVED FOR RELEASE: 06/20/2000**

Binary titanium-tin and titanium-zirconium alloys. S/762/61/000/000/002/029

Tests: (1) Three test specimens of each alloy were tested for tensile strength and elongation at room temperature. (2) Analogous tests at 400°C after 30-min soaking at test temperature. (3) Working ductility (formability): Bending angle for bending radius equal to sheet thickness at 20°C; minimal bending radius for a 90° bend; and stampability at 20 and 550-600°C. This testing method is described by the authors in Metallovedeniye i obrabotka metallov, no.3, 1959, 6-13. (4) Ductility of welds obtained by Ar-shielded automatic welding (fusion). Four specimens of each alloy (not heat-treated after welding) were tested for bending angle at 20°C (bending radius equal to sheet thickness). Details of the welding process are itemized. Test results with Ti-Sn alloys: (1) At 20°C the addition of up to 4% Zn improves the tensile strength only insignificantly above that of pure Ti (55 kg/mm²), but at 8-10% Sn the gain is appreciable (67-68 kg/mm²); at 400°C the strengthening effect is substantial: 34 kg/mm² for a 10%-Sn alloy, as against 22 kg/mm² for pure Ti. (2) Elongation has a distinct maximum at 1-3% Sn; the formability remains constant to 4-5% Sn and decreases with increasing % Sn. (3) Good weldability and elevated ductility up to 6% Sn; bend angle at a radius equal to sheet thickness: 100-110°. Conclusion: Sn - alloying has a beneficial effect on the formability and weldability of Ti-alloy sheets. Test results with Ti-Zr alloys: (1) At 20°C a 2% addition of Zr reduces the tensile strength of pure Ti by 3 kg/mm², an 8% Zr addition increases it by a like amount; at 4000 the tensile strength is increased from the

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Binary titanium-tin and titanium-zirconium alloys. S/762/61/000/000/002/029

22 kg/mm² of pure Ti to 32 kg/mm² with 8% Zr. (2) Elongation is greatest with 1-4% Zr (probably due to grain-size reduction as in the Ti-Sn alloy) and remains better than that of pure Ti even with 6-8% Zr. (3) Formability is not affected by up to 6% Zr, but decreases with greater amounts of Zr. (4) Weldability of alloys with 3-4% Zr was good; bend-test results were similar to those with Ti-Sn alloys. Conclusion: Zr is a desirable alloying element that improves the ductility, weldability, and high-temperature strength of Ti. There are 8 figures, 1 (unnumbered) table, and 4 references (1 Russian-language paper by authors, 3 English-language U.S.: Pietrokowsky, P., Frink, E.P., Trans. ASM, v.49, 1957, 339-358; Duwez, P., Inst. Met., J., v.80, no.9, 1952, 525; Finlay, W.L., et al., J. of Metals, v.6, 1954, 25).

ASSOCIATION: None given.

Card 3/3

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18.1285

5/129/61/000/004/007/012

E073/E535

AUTHORS: Luzhnikov, L.P., Candidate of Technical Sciences and

Novikova, V. M., Engineer

TITLE: Mechanical and Technological Properties of Ternary

Titanium Alloys

PERIODICAL: Metallovedeniye i termicheskaya obrabotka metallov, 1961, No.4, pp.31-35

TEXT: In an earlier paper of the authors (Ref.1) ternary titanium alloys were studied, most of which were based on the binary system Ti-Al with additions of one out of four β-stabilizers. In the here described work the system Ti-Sn was taken as the basis to which Zr, Cr, V, Mo and Mn were added. Furthermore, the system Ti-Al-Zr was studied. In all these ternary systems the alloys at the corner with 94% Ti, from 6% Sn and up to 6% of one of the enumerated β-stabilizers were studied. The alloys were manufactured in a vacuum arc furnace with a two-stage crystallizer with a weight of the melt of 3 kg. The ingots were forged and rolled under laboratory conditions. After sand-blasting and etching, sheets 1.3 to 1.5 mm thick were vacuum annealed for two hours at 800°C, followed by furnace cooling to 200°C and then

Card 1/6

Mechanical and Technological

89628 5/129/61/000/004/007/012 E073/E535

cooling in air. The compositions of thus produced sheets are All the alloys were subjected to short entered in Table 1. duration tensile tests at 20, 350 and 400°C. Following that the ductility, weldability and the microstructure were investigated. The strength and elongation of all the alloys at 20 and 400°C, except for the alloys of the system Ti-Sn-Mn, are plotted in Figs.1 and 2. The changes in the properties at 350°C are analogous to Fig.1 gives the strength ub, kg/mm² those at 400°C. elongation δ ,% at 20°C for the alloys Ti-Sn-Zr, Ti-Sn-Cr, Ti-Sn-Mo, Ti-Sn-V and Ti-Al-Zr (annealed sheets). Fig.2 gives the mechanical χ properties at 400°C for the same alloys. Fig. 3 gives the bending angles at 20°C for the same alloys (plot a - base metal, plot b weld joint produced by a tungsten electrode on a copper base in an argon atmosphere). It can be seen that the bending angle of the base metal in the systems Ti-Sn-Zr and Ti-Sn-V remain practically unchanged on transition from the tin-alloyed alloy to the alloy alloyed with V and Zr. In the system Ti-Sn-Mo and Ti-Sn-Cr, the bending angle decreases somewhat at first and then increases appreciably. In the system Ti-Al-Zr the bending angle increases Card 2/6

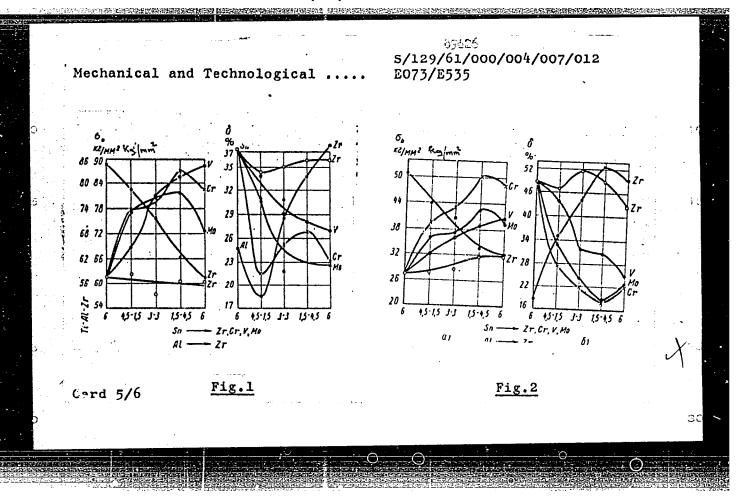
83:26 s/129/61/000/004/007/012 E073/E535

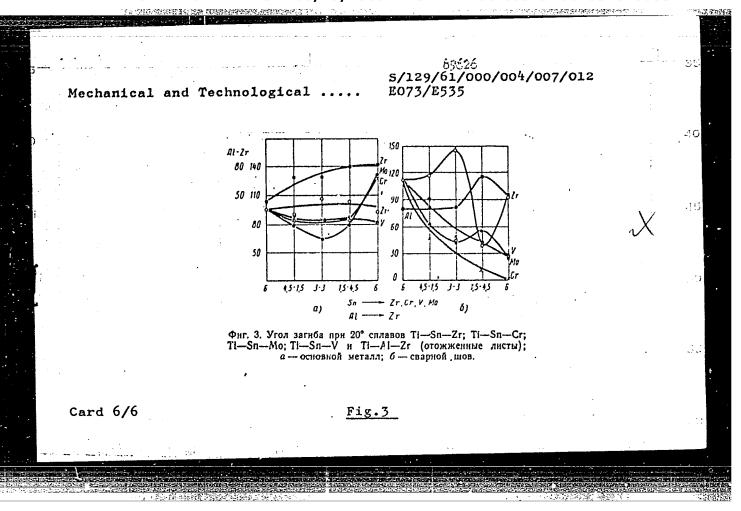
almost along a straight line on substituting aluminium alloying with zirconium alloying. The following conclusions are arrived at: 1. Substitution of aluminium by tin (within the limits of 6%) has no advantage for sheets which have to have a high strength at elevated temperatures, and satisfactory ductility and weldability. 2. Simultaneous alloying of titanium with aluminium or Zr or Sn and Zr permits obtaining alloys of a satisfactory strength, a very high ductility and good weldability. 3. The relations governing the changes in the basic properties in the case of combined alloying of Ti with Sn and one of the β -stabilizers are similar to the relations which were established for the case of simultaneous alloying with Al and one of the β -stabilizers. There are 5 figures, 2 tables and 3 references: all Soviet.

Card 3/6

Mechanical and Technological

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24193 5/129/61/000/007/007/016 E193/E135

AUTHORS:

Lazhnikov, L.P., Candidate of Technical Sciences, and

Moisevey, V. N., Engineer

TITLE

Alloys of the Titanium-Aluminium-Manganese system

FORTODICAL: Metallovedeniye i termicheskaya obrabotka metallov, 1961, No.7, pp. 29-34

TEXT: The object of the present investigation was systematically to study the mechanical and technological properties of the Ta-Al-In alloys with a view to determining the optimum composition of alloys of this type suitable for fabrication in the form or sheet and strip. The composition of the experimental alloys is given in Table 1 under the following headings: alloy number; chemical composition (%) Al, Mn (repeated three times). The experimental ingots were prepared from titanium sponge, A-00 grade aluminium, and Mr-1 grade manganese in an arc furnace with a consumable electrode in a mixture of argon and natium by the method of double smelting. The alloys contained the following impurities: 0.025-0.050% Se; 0.015-0.025% Si; 0.035-0.050% C; 0.050-0.070% O; 0.025-0.035% N; 0.006-0.009% H. The ingots were then rolled down Card 1/7

Alloys of the Ti-Al-Mn system

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to sheet 1.2-1.3 mm thick by the conventional methods, or forged to bars used for the preparation of impact strength test pieces. the first state of the investigation the following properties of the experimental alloys were determined. UTS at 20, 350 and 450°C; *longation at the same temperatures: impact strength at 20 $^{
m oC}$, dustility of the alloys and welded joints (determined by bending rests); deep-drawing characteristics at 20 and 600 $^{\circ}$ C. results which are reproduced graphically can be summarised as follows. 1) With increasing content of the alloying elements the WIS of titanium at room temperature increases from 51 to 115 kg/mm². its elongation decreases from 31 to 12%, and its impact strength decreases from 10 to 5.3 kgm/cm². 2) The UTS of titanium at elevated temperatures (350-450 °C) is increased by the additions of 3) The plasticity of the Ti-Al-Mn aluminium and manganese. alloys (as determined by the bending tests) decreases with increasing aluminium content and is increased by manganese additions of up to 9%. 4) Alloys with 6-7% of the alloying slements have satisfactory desp-drawing properties. At ejevated temperatures the Mn-rich alloys are better in this respect than the Al-rich materials. 5) The weldability of the Ti-Al-Mn Card 2/7

21,193

Alloys of the Ti-Al-Mn system

S/129/61/000/007/007/016 E193/E135

alloys (as determined by the ductility of argon-arc welded joins) depends on the manganese content and deteriorates as the composition of the altoy changes from the aluminium to the manganese side of the ternary Ti-Al-Mn constitution diagram. Alloys with 1.5-2% Mm (that is those consisting of the o-phase only) have best weldability, approaching that of pure titanium or Ti-Ai alloys. Alloys containing 5% or more manganese (that is those with the s + B structure) produce brittle welds. The austility of these brittle joins can be improved by aunealing at 750 °C. Welds anneated in this manner retained their dustility after 100 hours at 350, 450 and 550 °C with the exception of the ternary alloys containing more than 6.5% Al and the binary 9% Al-II allow. On the basis of these results 3 industrial Ti-Al-Mn alloys 074-1 (074-1) 074 (074), and BT4 (VT4) were developed, their allowing additions content being given below: 074-1, 1-2.5% Al and 0.8-2.0% Mm: 074, 2-3.5% Al and 1-2% Mm: VIL, 3.5-5% Al and 1-2% Mn. The maximum impurity content in all cases was: 0.4% Fer 0.15% Sir 0.1% Cr 0.15% Or 0.05% No 0.015% H. The mechanical properties of these alloys are given in Table 2, the first column of which reads: UIS, kg/mm2 vield card 3/ 7

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slongation in %; reduction in area, %: impact point in kg/mm²; strength, kg/mm2; Brinell bardness number, kg/mm2; elastic The strength of modulus, kg/mm2; snear modulus; Poisson ratio. these alloys decreases gradually with temperature and falls sharply above 450 °C which appears to be their maximum operating remperature. Their creep properties, satisfactory up to 350 °C. deteriorate at higher temperatures, which limits the field of application of these alloys. All these alloys can be readily argon-arc or spot-welded. The alloys are capable of being drawn, alloys OT4-1 and VT4 being, respectively, the most and least suitable for this purpose. Complex components can be drawn or pressed in several operations with intermediate anneals, or by preheating the blanks to 500-700 °C. The only heat treatment applicable to these alloys is annealing, the optimum annealing temperature increasing with increasing aluminium content. effect of annealing on mechanical properties of alloy OT4 is illustrated in Fig. 5, where UTS (σ_b , kg/mm², left-hand scale) and elongation (b. %, right-hand scale) of strip preliminarily coldrolled to 30% reduction in thickness, are plotted against the annealing temperature (OC), the duration of the annealing Card 4/7

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Alloys of the Ti-Al-Mn system

S/129/61/000/007/007/016 E193/E135

treatment being 30 minutes. The optimum annealing temperature for alloys OT4-1, OT4 and VT4 has been found to be 700-750.750-800, and 800-850 °C, respectively. In practice, however, owing to the tendency to excessive scale formation, lower annealing temperatures are employed (720-750 °C for alloy VT4, and 670-720 °C for alloys OT4 and OT4-1.). Internal stresses are removed by annealing at 500-600 °C. The annealing time is calculated on the basis of 10-15 minutes for each mm thickness of the strip, and should not exceed one hour. Acknowledgments are made to V.I. Dobatkin, I.N. Kaganovich, N.F. Anoshkin, S.A. Kudakevich and V.M. Novikova, who participated in this work.

There are 5 figures, 3 tables and 2 references: 1 Soviet and 1 English. The English language reference reads as follows:

Ref.1: R.F. Domagal, W. Rostoker. "TASM", V.47, 1955.

Card 5/7

LIVANOV, Vladimir Aleksandrovich; BUKHANOVA, Anna Arkhipovna; KOLACHEV, Boris Aleksandrovich; LUZHNIKOV, L.P., red.; ARKHANTEL'SKAYA, M.S., red. izd-va; DOBUZHINSKAYA, L.V., tekhn. red.

[Hydrogen in titanium]Vodorod v titane. Moskva, Metallurgizdat, 1962. 244 p. (MIRA 15:8) (Titanium-Hydrogen content)

S/129/63/000/002/003/014 E195/E383

AUTHORS: Luzhnikov, L.P., Novikova, V.M. and Mareyev, A.P.

TITLE: Solubility of the β-phase stabilizing elements in

 α -titanium

PERIODICAL: Metallovedoniye i termicheskaya obrabotka metallov, no. 2, 1963, 13 - 16

TEXT: The solid solubility of Fe, Cr, Mn, Si, Cu, Mo, V and Ta in α-Ti and in the α-phase of the Ti-6? Al alloy was studied by electrical resistivity and hardness measurements supplemented, when necessary, by X-ray diffraction analysis and metallographic examination. The results are reproduced in Tables 1 and 2. There are 6 figures and 2 tables.

Key to Table 1: 1 - alloy system; 2 - temperature, °C; 3 - time at temperature, hrs; 4 - solubility of the alloying element, %.

Card 1/3

Solubility of			5/129, E193/	/63/000/002/003/014 e383	
Table 1: Sol	ubility of a	ome elem	ents in α-	titanium'	
	Система спланов	Температура	Brews HL. Pactbook-		
	TI—Fe TI—Cr TI—Cr	500 650	150—125 125 100 150 0,2—0,3 0,3—0,5 0,4—0,6 0,4—0,6	3	
	TI-Mn TIMn TISI TISI	400 530 600700 840	125 0,5—0,7 100—7 0,3—0,8 50 0,5—0,1 125—100 0,4—0,6	7 5 7	
	T!—Cu T!—Cu T!—Mo T!—Mo	700 500—600 750	75 0,5—0, 125—100 0,3—0, 75 0,3—0, 125—75 0,5—1,	7 5 4	
	Ti—V Ti—Ta Ti—Ta	500—600— 750 500 700	125 6,5—8 75 5—6 50 4—5		•
	Ti-Ta	800	W 1-0	-	
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Solubility of

Table 2: Solubility of some elements in the α-phase of Ti-6% Al alloy.

Key - 1 - alloy system; 2 - temperature, °C; 3 - time at temperature, hrs; 4 - solubility of the alloying element, %.

Система силавов	Температура в °С	Bpens narpesa B 4	Раствори- мость в %
Ti_Al_Fe Ti_Al_Fe Ti_Al_Fe Ti_Al_Cr Ti_Al_Cr Ti_Al_Mn Ti_Al_Mn Ti_Al_Mn Ti_Al_Si Ti_Al_Cu Ti_Al_Mo Ti_Al_Mo Ti_Al_Mo Ti_Al_Mo Ti_Al_Mo	600—700 800 600 700—800 500—850 600—700—800 750—850 600—750—850	125 100—75 125—100 75 125 125—75 125—50 125—100—75 125—50 125—50 125—75—50 125—75—50	0,2-0,4 0,2-0,4 0,5-1,2

Card 3/3

KOLOBNEV, Ivan Filippovich; LUZHNIKOV, L.P., red.; MISHARINA, K.D., red.izd-va; KARASEV, A.I., tekhn. red.

[Heat-resistance of aluminum foundry alloys] Zharoprochnost' liteinykh aliuminievykh splavov. Moskva, Metallurgizdat, 1964. 223 p. (MIRA 17:3)

B

L 14319-65 ENT(m)/EMP(b)/EMA(d)/EMP(w)/EMP(t) ASD(m)-3/AFETR/IJP(d) ACCESSION NR: AT4048055 MJW/JD/MLK S/0000/64/000/000/0080/0087

AUTHOR Luzhníkov, L.P., Novikova, V.M., Mareyev, A.P.

TITLE: Dilatometric studies of <u>transformations</u> in <u>titanium alloys</u> 1名 ンク

SOURCE: Soveshchaniye po metallurgii, metallovedeniyu i primeneniyu titana i yego splavov. 5th, Moscow, 1963. Metallovedeniye titana (Metallography of titanium); trudy* soveshchaniya. Moscow, Izd-vo Nauka, 1964, 80-87

TOPIC TAGS: dilatometry, titanium alloy, alloy phase transformation, titanium mechanical property, titanium microstructure, titanium alloy aging

ABSTRACT: This is part of an extended study on the reaction of titanium alloys to thermal treatment, including a study of mechanical properties and microstructure. Dilatometric aging and surface hardening studies were conducted on the industrial alloys YT3-1, VT8 VT9 and VT14 containing add tions of 2 to 4 of the elements Al, alloys YT3-1, VT8 NT9 and Sn in various amounts, as well as on the systems Ti-Cr, Ti-Mo, Cr, Mo, V, Si and Sn in various amounts, as well as on the systems Ti-Cr, Ti-Mo, Ti-Al-Cr and Ti-Al-Mo, containing 3-10% of these additions, at temperatures increasing up to 1000C by 4.5-5 degrees/min. Dilatometric curves are presented for the various alloys and related to their microstructures and phases. The effect of 10 hours' aging on titanium alloys is graphed and the relation between mechanical properties and the

L 14319-65 ACCESSION NR: AT4048055

length of aging tabulated for VT3-1. No dilatometric effect was seen in the industrial alloys upon heading from the quenched state, with the exception of VT3-1 (5%Al, 1.7% alloys upon heading from the quenched state, with the exception of VT3-1 (5%Al, 1.7% Cr, 2% Mo) which showed a negative effect in the 285-400C range after quenching from the two-phase range (845C). This points toward the appearance of the ω -phase upon dissociation of the β -phase. A negative dilatometric effect (ω -phase) was also seen in the experimental alloys at temperatures slightly below or above those for the in the experimental alloys at temperatures slightly below or above those for the VT3-1 ω -phase. The correlation of phase and hardness (Pockwell test) is discussed. VT3-1 ω -phase. The correlation of phase and hardness was found at the For the Ti-Cr alloys and VT3-1, for example, the maximal hardness was found at the β -phase containing Cr and Mo is fixated in supercritical concentration. The ω -phase will appear as a result of aging in the 350-400C range. The amount of fixed β -phase will appear as a result of aging in the 350-400C range. The amount of fixed β -phase may apparently reach 30-40%, that of the ω -phase 50%. This explains the considerable effect of aging on properties of the VT3-1 alloy. Orig. art. has: 8 figures and 2 tables.

ASSOCIATION: none

SUBMITTED: 15Jul64

ENCL: 00

SUB CODE: MM

NO REF SOV: 006

OTHER: 001

Card 2/2

S/2981/64/000/003/0209/0215

ACCESSION NR: AT4037662

AUTHOR: Luzhnikov, L. P.

TITLE: Significance of silicon in type AK4 aluminum alloys (group RR)

SOURCE: Alyuminiyevy*ye splavy*, no. 3, 1964. Deformiruyemy*ye splavy* (Malleable alloys), 209-215

TOPIC TAGS: aluminum alloy, alloy AK4, alloy AK4-1, alloy RR58, alloy RR59, RR alloy group, alloy mechanical property, alloy heat resistance, heat resistant alloy, silicon admixture

ABSTRACT: The content of Si was varied from 0.03 to 2.8% in ingots (diameter 70 mm) of alloys AK4 and AK4-1, which correspond to alloys RR58 and RR59 as developed by Rolls Royce, England. Other alloying elements were held constant (2.05-2.10% Cu, 1.6-1.76% Mg, 1.24-1.26% Ni, 1.32-1.37% Fe). Rods (diameter 18 mm) were pressed at 420 C, final Mg, 1.24-1.26% Ni, 1.32-1.37% Fe). Rods (diameter 18 mm) were pressed at 420 C, final Mg, 1.24-1.26% Ni, 1.32-1.37% Fe). Rods (diameter 18 mm) were pressed at 420 C, final Mg, 1.24-1.26% Ni, 1.32-1.37% Fe). Rods (diameter 18 mm) were pressed at 420 C, final Mg, 1.24-1.26% Ni, 1.32-1.37% Fe). Rods (diameter 18 mm) were pressed at 420 C, final Mg, 1.24-1.26% Ni, 1.32-1.37% Fe). Rods (diameter 18 mm) were pressed at 420 C, final Mg, 1.24-1.26% Ni, 1.32-1.37% Fe). Rods (diameter 18 mm) were pressed at 420 C, final Mg, 1.24-1.26% Ni, 1.32-1.37% Fe). Rods (diameter 18 mm) were pressed at 420 C, final Mg, 1.24-1.26% Ni, 1.32-1.37% Fe). Rods (diameter 18 mm) were pressed at 420 C, final Mg, 1.24-1.26% Ni, 1.32-1.37% Fe). Rods (diameter 18 mm) were pressed at 420 C, final Mg, 1.24-1.26% Ni, 1.32-1.37% Fe). Rods (diameter 18 mm) were pressed at 420 C, final Mg, 1.24-1.26% Ni, 1.32-1.37% Fe). Rods (diameter 18 mm) were pressed at 420 C, final Mg, 1.24-1.26% Ni, 1.32-1.37% Fe). Rods (diameter 18 mm) were pressed at 420 C, final Mg, 1.24-1.26% Ni, 1.32-1.37% Fe). Rods (diameter 18 mm) were pressed at 420 C, final Mg, 1.24-1.26% Ni, 1.32-1.37% Fe). Rods (diameter 18 mm) were pressed at 420 C, final Mg, 1.24-1.26% Ni, 1.32-1.37% Fe). Rods (diameter 18 mm) were pressed at 420 C, final Mg, 1.24-1.26% Ni, 1.32-1.37% Fe). Rods (diameter 18 mm) were pressed at 420 C, final Mg, 1.24-1.26% Ni, 1.32-1.37% Fe). Rods (diameter 18 mm) were pressed at 420 C, final Mg, 1.24-1.26% Ni, 1.32-1.37% N

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ACCESSION NR: AT4037662

this is related to complex changes in phase composition of the alloys. A sharp peak in stress-rupture strength and creep strength at high temperature was noted for 1.3 to 1.8% Si. The effects noted for Si are presumed also to apply to other alloys in the system Al-Cu-Mg-Si. "V. M. Novikova, V. F. Murzova and A. P. Mareyev took part in the work." Orig. art. has: 3 graphs.

ASSOCIATION: none

SUBMITTED: 00

DATE ACQ: 04Jun64.

ENCL: 02

SUB CODE: MM

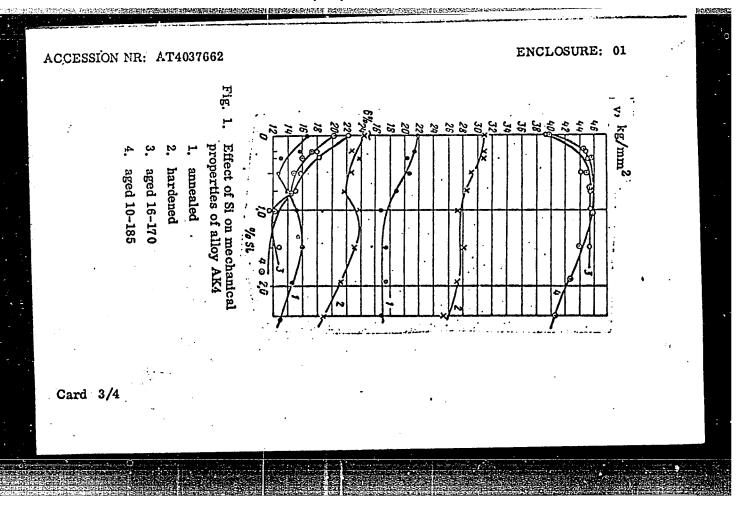
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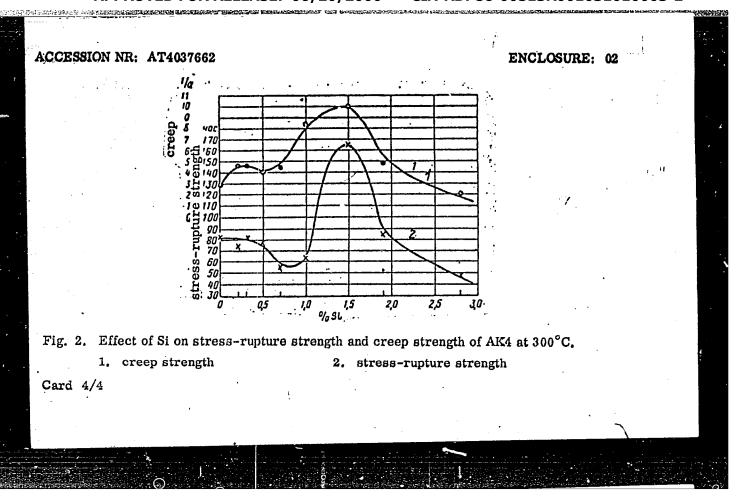
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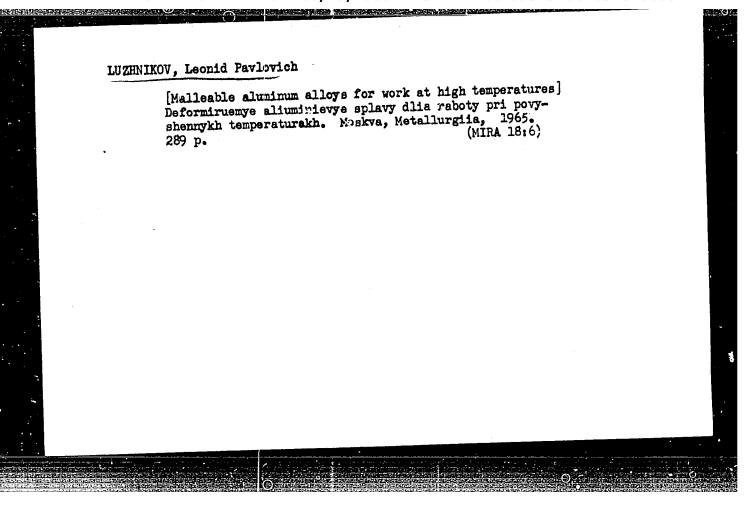
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TITLE: The effec	ts of <u>leat treatment</u> on edeniye i termicheskaya	M.; Mareyev, A. P.; Orlova, I. S transformations in Ti alloys obrabotka metallov, no. 5, 1965,	33 32 3 8 21-28
expansion ABSTRACT: Variou to ascertain the latometric sample sults are given i from various tem 750 to 1050°C. I place in aged con	to 10 conditions for w-phase as were made, and appropriate form of dilatomet peratures and aged at 35 it was found that transfumercial VT3-1 alloy. A	ent, dilatometry, coefficient of the variation in matastable β-phase a riate experiments were completed. The quench temperatures ran ormation of the metastable β-phase fter aging at temperatures in the and for the range 350-450°C, a ormations also takes place in the comparisons are comparisons at the comparisons also takes place in the comparisons are comparisons at the comparisons at the comparisons at the comparisons are comparisons at the comparisons	ed in ord lloys. Di The re- les quenc ged from e takes 350-370°
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CCESSION NR: AP50131	58			:::::::::::/::
after quenching from the limited to the use of ling, the authors give the effects of any phatche hardness curves, rong, art. has: 7 fig	the alloys in juxtaposed by se changes on elating the a	n the quenched con princes and ditato n strength propert appearance of w-ph	dition (without a metry curves, in ies:()Maxima were	ging). In clo order to show observed in
ASSOCIATION: none				
SUBMITTED: 00		ENCL: 00	SUB, CC	de: MM
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		669.295	33 25
AUTHOR: Luzhniko	r, L. F.; Novikova, V. M.	; Mareyev, A. P.	$\boldsymbol{\beta}$
TITLE: Hardenabi	lity of commercial Ti all	Loys	
"你 我一个女人,我们有一个人的人,我们就是什么		obrabotka metallov, no. 5	, 1965 , 53–56
ABSTRACT: The hamined. The alloy end hardenabilitiing and aging; the treatment schedul and sometimes threatmy are given grand hardness was presented.	s were processed into ro- es were determined by the e specimens were section e after the quench consi- ee separate temperatures raphically for the eight lotted against distance	ifferent commercial Ti alds of diameters ranging a standard end-quench mered and hardnesses determined of heating the samp for one hour each. The alloys ari their respections the quinched surfactions were observed in soing hardness with distance	hod. After quench- ined. The heat les at one, two, results of the tive treatments. e, for quenched, me cases, the most

L 57507-65 ACCESSION NR: AP5013162		
【B	ness drops more slowly. Hardnes of the alloys upon heat treatmen	Ca mid cure of an area
groups were distinguishable	: a) alloys quenched rapidly to	for VT18 VT9 and VT9-1
b) alloys quenched to form and VT16), c) alloys, in wh	ich the 8-phase is fixed upon qu	0.000
art. has: 1 figure, 1 tabl	e.	
ASSOCIATION: none		
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